

## Powder X-ray Diffraction

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## Uses of Powder Diffraction

### Qualitative Analysis

Identification of single-phase materials  
Identification of multiple phases in microcrystalline mixtures  
Recognition of amorphous materials in partially crystalline mixtures

### Structure Refinement

Rietveld Method

### Structure Solution

Reciprocal Space Methods  
Real Space Methods

### Thermal expansion and Phase Transitions

### Quantitative Analysis

Lattice Parameter Determination  
Phase Fraction Analysis

### Peak Shape Analysis

Crystallite Size Distribution  
Microstrain Analysis  
Extended Defect Concentration

## Three Unique Features of Synchrotron Radiation

### •Intensity

#### •Enables Rapid Data Collection

Kinetics  
Unstable Compounds  
Environmental Cells  
–Enables Focussing  
Small Samples  
Small areas/volumes

### •Energy Range

#### •Enables Spectroscopy

–Elemental Identification  
–Bonding Studies  
–Speciation

#### •Enables Optimal Conditions

–Environmental Cells  
–Selected Elements

### •Low Divergence

Enables High Resolution

•Micro Beams  
•Small Volumes  
•Complex Materials

## What is special about a crystal?

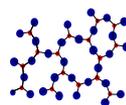
*Solid phases are often crystalline, but need not be - e.g. glass an "amorphous material"*

### Glass

- Fractures into shards
- Takes on any shape, depending on preparation
- Properties do not vary with orientation.

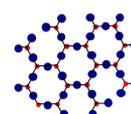
### Crystal

- Cleaves along preferred directions
- Grows with well developed crystal faces
- Properties depend on orientation in which they are measured.



•Si

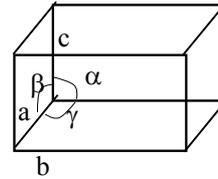
•Oxygen



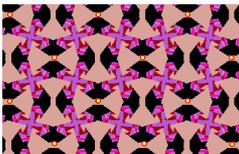
## Crystal Structure

- **CRYSTAL:** Contains a periodical array of atoms/ions. This can be represented by a simple lattice of points.
- A group of atoms is associated with each lattice points.
- **LATTICE:** An infinite array of points in space, in which each point has identical surroundings to all others.
- **CRYSTAL STRUCTURE:** The periodic arrangement of atoms in the crystal.

## The Unit Cell



The unit cell is a basic parallelepiped shaped block from which the whole volume of the crystal may be built by repetition in 3 dimensions. Any point in the unit cell may be specified with respect to the origin by parameters  $x, y, z$  measured parallel to the unit cell axes and expressed as fractions.

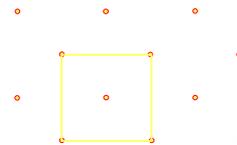


### Example of 2D symmetry in a wallpaper pattern

To show symmetry:

1. Pick a point
2. Find all equivalent points

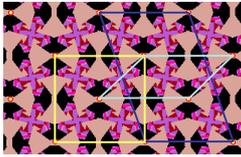
(<http://www.clarku.edu/~djoyce/wallpaper/>)



### Example of 2D symmetry in a wallpaper pattern

To show symmetry:

- .1. Pick a point
  - .2. Find all equivalent points
- .These points form a 2D lattice
- .Connecting 4 lattice points to form a parallelogram gives a possible *unit cell*



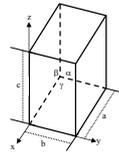
**Example of 2D symmetry in a wallpaper pattern**

. Connecting 4 lattice points to form a parallelogram gives a possible *unit cell*

. *Unit cell* – the basic unit that repeats in every direction

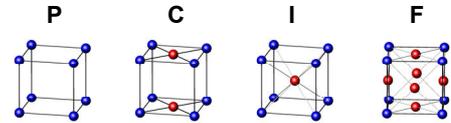
. Different *unit cells* can be chosen

.But some *unit cells* are preferable for higher symmetry

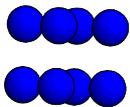


Lattice parameters:  
 $a, b, c; \alpha, \beta, \gamma$

Name	Bravais Lattice	Conditions
Triclinic	1 (P)	$a \neq b \neq c$ $\alpha \neq \beta \neq \gamma$
Monoclinic	2 (P, C)	$a \neq b \neq c$ $\alpha = \beta = \gamma \neq 90^\circ$
Orthorhombic	4 (P, F, I, A)	$a \neq b \neq c$ $\alpha = \beta = \gamma = 90^\circ$
Tetragonal	2 (P, I)	$a = b \neq c$ $\alpha = \beta = \gamma = 90^\circ$
Cubic	3 (P, F, I)	$a = b = c$ $\alpha = \beta = \gamma = 90^\circ$
Trigonal	1 (P)	$a = b = c$ $\alpha = \beta = \gamma < 120^\circ \neq 90^\circ$
Hexagonal	1 (P)	$a = b \neq c$ $\alpha = \beta = 90^\circ$ $\gamma = 120^\circ$



**PCC Lattice**



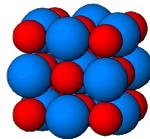
$\alpha$ -Po is **primitive-Centered Cubic**  
Identical atoms at corners but nothing at the and body or face centers.  
**Lattice type P**

**BCC Lattice**



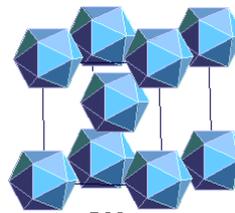
$\alpha$ -Iron is **Body-Centered Cubic**  
Identical atoms at corners and body center (nothing at face centers)  
**Lattice type I**  
Also Nb, Ta, Ba, Mo...

**FCC Lattice**



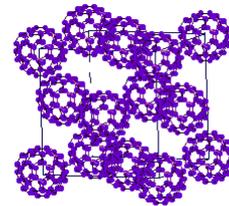
Sodium Chloride (NaCl) Na is much smaller than Cl  
**Face Centered Cubic**  
*Rocksalt structure*  
**Lattice type F**  
Also NaF, KBr, MgO....

**FOOT & MOUTH VIRUS**



BCC

**BUCKMINSTERFULLERENE**

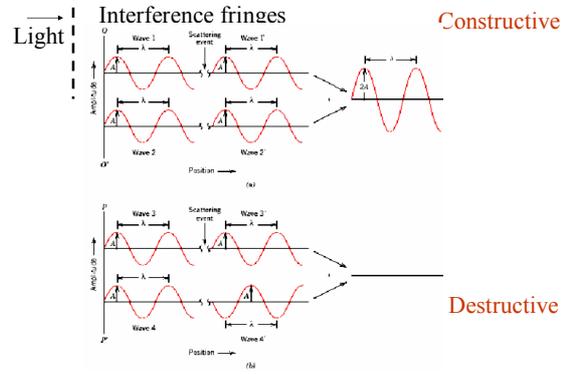
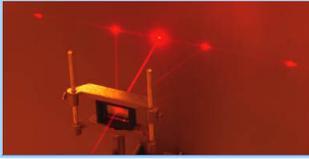


FCC

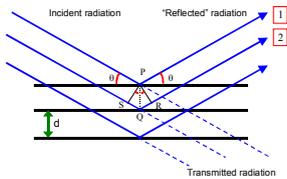
## Diffraction of X-ray Waves

- **Diffraction:** When light passes sharp edges or goes through narrow slits the rays are deflected and produce fringes of light and dark bands.

Diffraction grating and helium-neon laser



## Bragg's Law



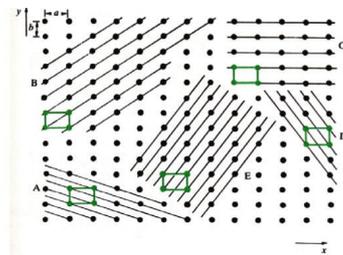
Beam "2" travels the extra distance SQR

$$\begin{aligned}
 n\lambda &= \overline{SQ} + \overline{QR} \\
 &= d_{hkl} \sin \theta + d_{hkl} \sin \theta \\
 &= 2d_{hkl} \sin \theta
 \end{aligned}$$

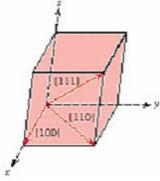
*But not all planes result in diffraction !!!*

## Lattice Planes

- It is possible to describe certain directions and planes with respect to the crystal lattice using a set of integers referred to as Miller Indices



## Crystallographic Directions And Planes



### Lattice Directions

Individual directions:  $[uvw]$

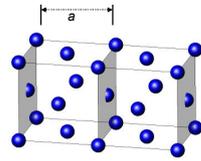
Symmetry-related directions:  $\langle uvw \rangle$

### Miller Indices:

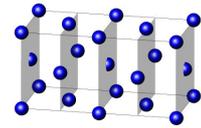
1. Find the intercepts on the axes in terms of the lattice constant a, b, c
2. Take the reciprocals of these numbers, reduce to the three integers having the same ratio  
(hkl)

Set of symmetry-related planes:  $\{hkl\}$

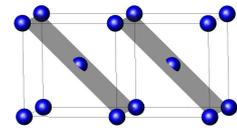
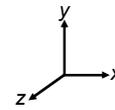
## Examples of Miller Indices



(100)



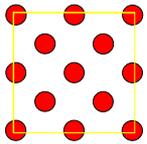
(200)



(110)

## Families of Planes

- Miller indices describe the orientation of a family of planes
  - the spacing between adjacent planes in a family is referred to as a “d-spacing”
- different families of planes
  - d-spacing between (400) planes is 1/4 that of the (100) spacing.
  - The (300) plane does not contain atoms and so is not observed

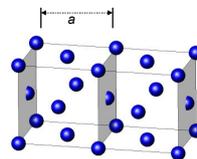


_____	_____	_____	.....
_____	_____	_____	.....
_____	_____	_____	.....
_____	_____	_____	.....
(100)	(200)	(400)	(300)

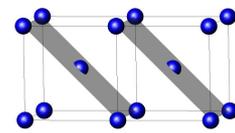
## Lattice Spacing

$$\frac{1}{d_{hkl}^2} = \frac{h^2 + k^2 + l^2}{a^2}$$

For cubic system with  $a = 4.0 \text{ \AA}$

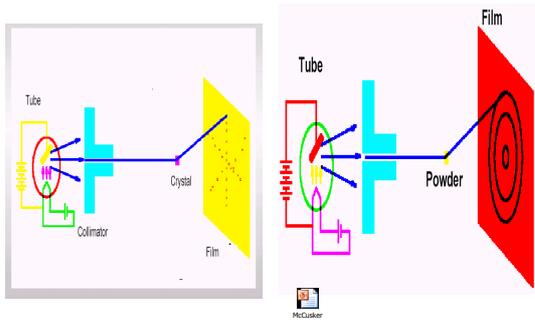


$d_{100} = 4.0$

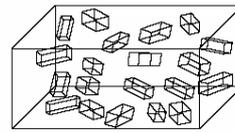


$d_{110} = 2.828$

## Single Crystal vs Powder

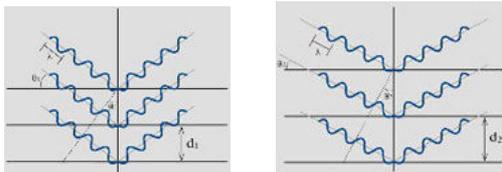
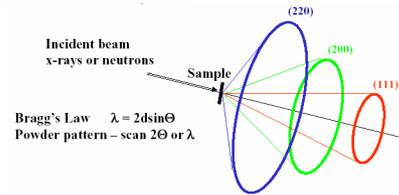


## Powder – A Polycrystalline Mass



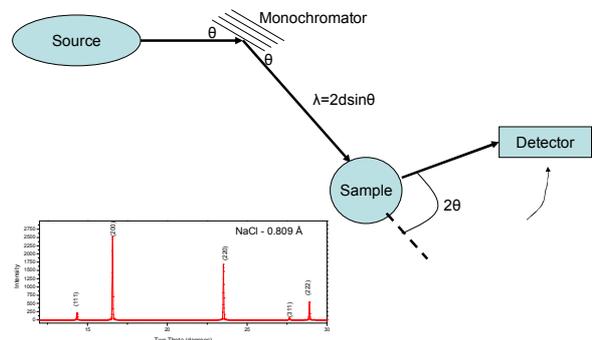
All orientations of crystallites possible

Single crystal reciprocal lattice - smeared into spherical shells



- By varying the angle  $\theta$ , the Bragg's Law conditions are satisfied by different  $d$ -spacings in polycrystalline materials.
- Plotting the angular positions and intensities of the resultant diffracted peaks produces a pattern which is characteristic of the sample.

## Powder Diffraction



## Information Contained in a Diffraction Pattern

### Peak Positions

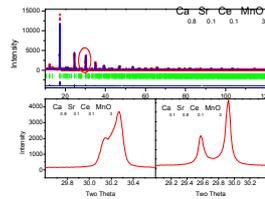
Crystal System  
Space Group Symmetry  
Unit Cell Dimensions  
Qualitative Phase Identification

### Peak Intensities

Unit Cell Contents  
Point Symmetry  
Quantitative Phase Fractions

### Peak Shapes & Widths

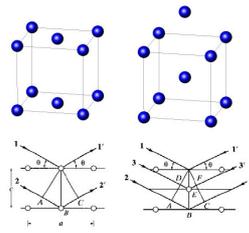
Crystallite Size (2-200 nm)  
Non-uniform microstrain  
Extended Defects (stacking faults, etc.)



Changes in symmetry and microstrain upon chemical substitution can be established by examination of the patterns

## Centering and Absences

- The positions of the atoms in a unit cell determine the intensities of the reflections
- Consider diffraction from (100) planes in

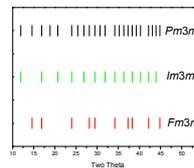
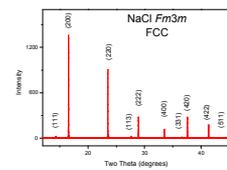
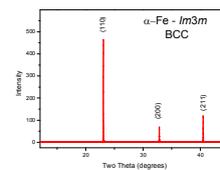


If the pathlength between rays 1 and 2 differs by  $\lambda$ , the path length between rays 1 and 3 will differ by  $\lambda/2$  and destructive interference in (b) will lead to no diffracted intensity

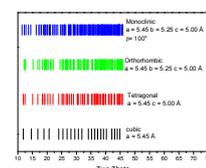
## Centering and Absences

- We can extend these types of calculation to include other modes of lattice centering. They all lead to systematic absences

Bravais lattice	Reflections that must be absent
Simple (Primitive)	none
Base (C) centered	h and k mixed
Body (I) centered	(h+k+l) odd
Face (F) centered	h, k and l mixed



Influence of centering



Influence of symmetry

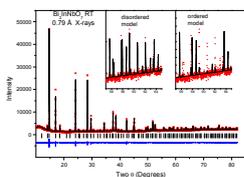
## Multiplicity

- For high symmetry materials the Bragg angles and d-spacings for different reflections may be equivalent to one another  
For example (100), (010), (001) etc are equivalent in a cubic material
- In a powder, all planes with the same d-spacing contribute to the scattered intensity at a given Bragg angle
- The number of planes that are symmetry equivalent is referred to as the multiplicity and its appears as a multiplicative term in powder diffraction intensity calculations
- The multiplicity of a reflection depends upon the symmetry of the crystal  
Multiplicity of {100} for cubic is 6, but for tetragonal it would only be 4 as (100) and (001) are not equivalent

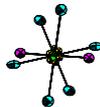
## Diffraction Patterns

- Spacing of peaks depends on size of unit cell and the space group.
- The bigger the unit cell and/or the lower the symmetry the more diffraction peaks are observed.
- Intensity of peaks depends on (amongst other things) the arrangement of the atoms in the unit cell.
- For two materials that had identical unit cells, the peak positions would be IDENTICAL, however their intensities would be DIFFERENT.

## Need for High Q



There are many more reflections at higher Q. Therefore, most of the structural information is at higher Q



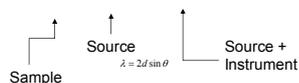
Refinement of structure gave unusual displacement parameters for the Bi cations, indicative of cation disorder. The patterns could only be adequately fitted by including 6-fold disorder of the Bi. This involves a displacement along the (1 -1 0) direction

Atom	Site	x	y	z	B <sub>iso</sub>
<b>Model 1. Ordered Bi, R<sub>w</sub> 4.98 R<sub>p</sub> 6.97%</b>					
Bi	16d	0	0.25	0.75	2.74(6)
In/Nb	16c	0	0	0	3.00(8)
O(1)	48f	0.350(3)	0.125	0.125	7.6(7)
O(2)	8b	0.375	0.375	0.375	7.6(7)
<b>Model 2. Disordered Bi, R<sub>w</sub> 3.09 R<sub>p</sub> 3.93%</b>					
Bi	96h	0	0.2249(1)	0.7751(1)	0.98(7)
In/Nb	16c	0	0	0	0.61(3)
O(1)	48f	0.322(1)	0.125	0.125	1.7(2)
O(2)	8b	0.375	0.375	0.375	1.7(2)

## Need for High Resolution

$$\frac{\Delta d}{d} = \frac{\Delta \lambda}{\lambda} + \frac{\Delta \theta}{\tan \theta}$$

Differentiating Bragg's Law gives the resolution as:

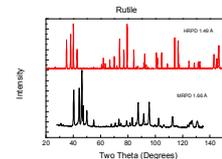


### Resolution

In Powder Diffraction it typically refers to the width of the peaks.

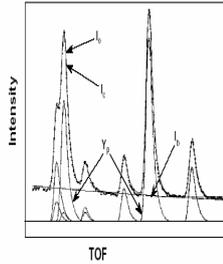
In Single Crystal Diffraction it typically refers to the minimum d-space studied.

Both definitions are relevant.



## Peak Overlap

- Powder Diffraction patterns are a one dimensional representation of a three dimensional structure.
- Often peaks due to individual Bragg reflections overlap



## The Solution - Rietveld

$$y_{\text{icalc}} = y_{\text{iback}} + \sum_p \sum_{k=k_1^p}^{k_2^p} G_{ik}^p I_k^2$$

- $y_{ic}$  the net intensity calculated at point  $i$  in the pattern,
- $y_{\text{iback}}$  is the background intensity,
- $G_{ik}$  is a normalised peak profile function,
- $I_k$  is the intensity of the  $k^{\text{th}}$  Bragg reflection,
- $k_1 \dots k_2$  are the reflections contributing intensity to point  $i$ ,
- the superscript  $p$  corresponds to the possible phases present in the sample.

## The Answers

- The Profile R

$$R_p = \frac{\sum |y_{\text{iobs}} - y_{\text{icalc}}|}{\sum y_{\text{iobs}}}$$

- The weighted Profile R

$$R_{\text{wp}} = \left[ \frac{\sum w_i (y_{\text{iobs}} - y_{\text{icalc}})^2}{\sum w_i y_{\text{iobs}}^2} \right]^{1/2}$$

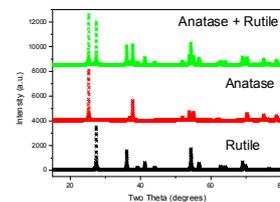
- The expected Profile R

$$R_{\text{exp}} = \left[ \frac{N - P}{\sum w_i y_{\text{iobs}}^2} \right]^{1/2}$$

- The Goodness of fit

$$\chi^2 = \frac{\sum w_i (y_{\text{iobs}} - y_{\text{icalc}})^2}{N - P} = \left[ \frac{R_{\text{wp}}}{R_{\text{exp}}} \right]^2$$

## Phase Analysis



- Where a mixture of different phases is present, the resultant diffraction pattern is formed by addition of the individual patterns.
- The intensity of the peaks is proportional to the amount of the phase present.

## Quantitative Phase Analysis

- Bragg scattering is proportional to  $N/V$  where  $N$  is the number of unit cells and  $V$  the unit cell volume. There for the weight of a phase in the beam is:

$$W_p = \frac{(SZMV)_p}{\sum_i (SMPV)_i}$$

S - the scale factor

Z the number of formula unites per unit cell

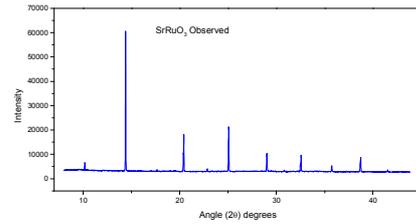
M the molecular weight of the formula unit

I is the index running over all phases

- Hence SZVM is proportional to the weight of the diffracting sample

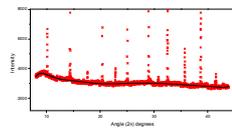
## An Example

- Synchrotron X-ray Diffraction pattern for  $\text{SrRuO}_3$



## The background

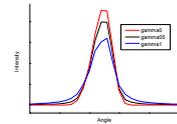
- Fluorescent radiation from the sample
- Diffraction from the continuous spectrum
- Diffuse scattering
  - Incoherent
  - Temperature diffuse
- Other materials
  - Specium holder
  - air etc



- Background can be either fitted or estimated.
- Here the capillary is a feature.

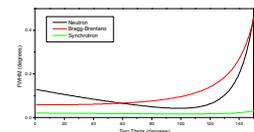
## Peak Shapes

- Different Diffractometers have different peak shapes.
- The most widely function is a pseudo-Voigt (mixed Gaussian and Lorentzian).



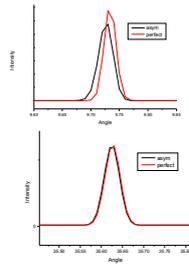
$$G_{ik} = \gamma \frac{C_0^{1/2}}{H_k \pi} [1 + C_0 X_{ik}^2]^{-1} + (1 + \gamma) \frac{C_1^{1/2}}{H_k \pi^{1/2}} \exp[-C_1 X_{ik}^2]$$

- The width of peaks is usually not constant.
- $$H^2 = U \tan^2 \theta + V \tan \theta + W$$



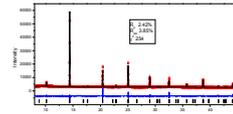
## Peak Asymmetry

- Beam Divergence can result in asymmetric peaks at low angles.
- Results from not integrating over the entire Debye cone.



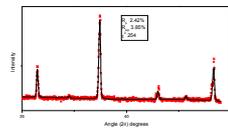
## The Simple Structural Model

- The fit to a single phase sample looks good



- BUT.....

- The detail of the fit is not satisfactory - the model is missing something!

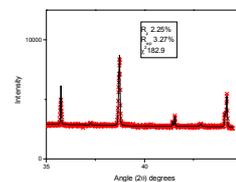


## A Common Problem

- If the structural model is wrong then the most common response of Rietveld programs is to:
  - broaden the peaks,
  - Increase the displacement parameters,
- The former is most noticeable at high angles where intensity is lowest.
- Due to absorption of the X-rays powder X-ray diffraction often yields poor displacement parameters

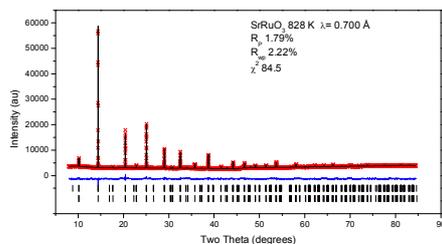
## An Alternate Model

- The high angle splitting is well modeled by a tetragonal model - but this overestimates some intensities.



- The Truth lies somewhere in the middle

## The finished Product



- The sample contains a mixture of both phases!

## Strengths and Limitations of Powder X-ray Diffraction

### Strengths

- Non-destructive – small amount of sample
- Relatively rapid
- Identification of compounds / phases – not just elements
- Quantification of concentration of phases – (sometimes)
- Classically for powders, but solids possible too
- Gives information regarding crystallinity, strain, crystallite size, and orientation

### Limitations

- Bulk technique – generally – unless a microfocus source is used
- Not a “stand-alone” technique – often need chemical data
- Complicated appearance
- multiphase materials – identification /quantification can be difficult

## Experiment Design Issues

### *What Wavelength?*

- Absorption is your enemy!
- Short Wavelengths are best! BUT....
- Consider required resolution. And
- Avoid Absorption Edges.

### *What Size Capillary?*

- Small capillaries reduce absorption AND improve resolution.
- BUT reduce amount of material.